

Magnetic materials for nano-scale sensors

Bob O'Handley, M.I.T.

Objectives:

- | | |
|------------------------|--------------------------------|
| Low power consumption | □ Smaller |
| Enhanced sensitivity | □ lower noise, no domain walls |
| Enhanced functionality | □ novel couplings |

With strong reference to work by:

Caroline Ross (M.I.T.),	Mike McHenry (C.M.U.),
Giesle Herzer (VAC/Morgan),	Manfred Wuttig (U. Md),
Francesco Stellacci (M.I.T.),	Moungi Bawendi (M.I.T.)

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Materials matrix suggested by Workshop

Outline

- 1) **Functionality:** beyond flux, Hall, MR...
- 2) **Materials suggestions (beyond single-phase thin films):**
 - Not new compositions; new materials:
Nano-structured magnetic materials,
Immiscible materials
 - Relevant length scales and
implications for sensor materials
- 3) **Processing opportunities**
 - From Top-down lithography
of nano-magnetic structures
to Bottom-up, Self assembled nano-magnetics,
immiscible systems; metallic...polymeric
Carbon-based (Fullerenes, nanotubes)

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1) Magnetic material functionality

The obvious:

- Induction
- flux channel
- Hall effect
- MR, GMR, tunnel MR
- Magneto-elastic
- Magneto-optic

- dB/dt sensor
- Shield or focus H
- H sensor
- H sensor
- Accelerometer
- Optical rotation
(or inverse?)

Perhaps less obvious:

Magneto caloric
Magneto-electric
Spin-semiconductor hetero-effects

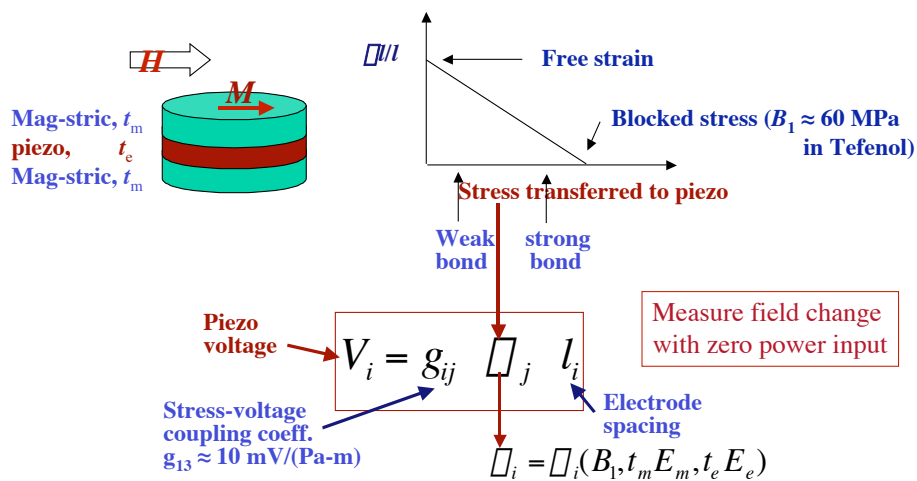
H sensor or accelerometer

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Magnetostrictive/electro-active sensors



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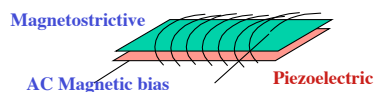
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Prior art: magnetostrictive/electro-active field sensors

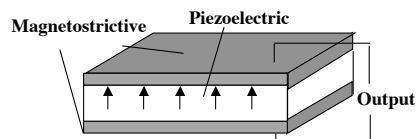
Marconi Research d_{31} sensor

Single magnetic layer,
AC bias field on magnetic layer,
Lynch and Gallantree. GEC Jnl. Of Res. **8**, (1990)



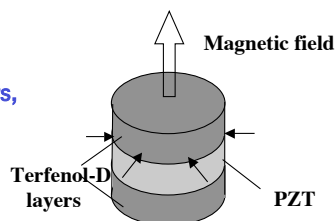
Spinix d_{31} sensor

Two or more magnetic layers,
Device is **passive**, no power needed,
Y-Q. Li and R. C. O'Handley,
J. Appl. Sens. Tech. **17**, 10 (2000)



Penn State sensor

Two compressively prestressed Terfenol-D layers,
Device is **passive**, no power needed,
J. Ryu, A. V. Carazo, K. Uchino, and H. -E. Kim,
Jpn. J. Appl. Phys. **40**, 4948 (2001)

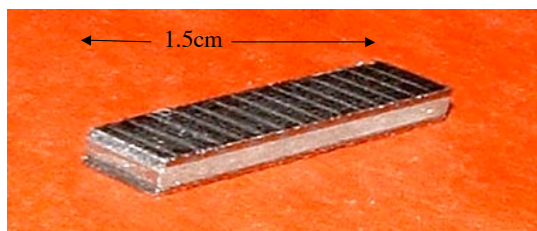


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New magnetostrictive/electro-active field sensor



Ferro Solutions sensor

Two Terfenol-D layers, PZT
Device is **passive**, no power needed: sensitivity = 600 nV/nT

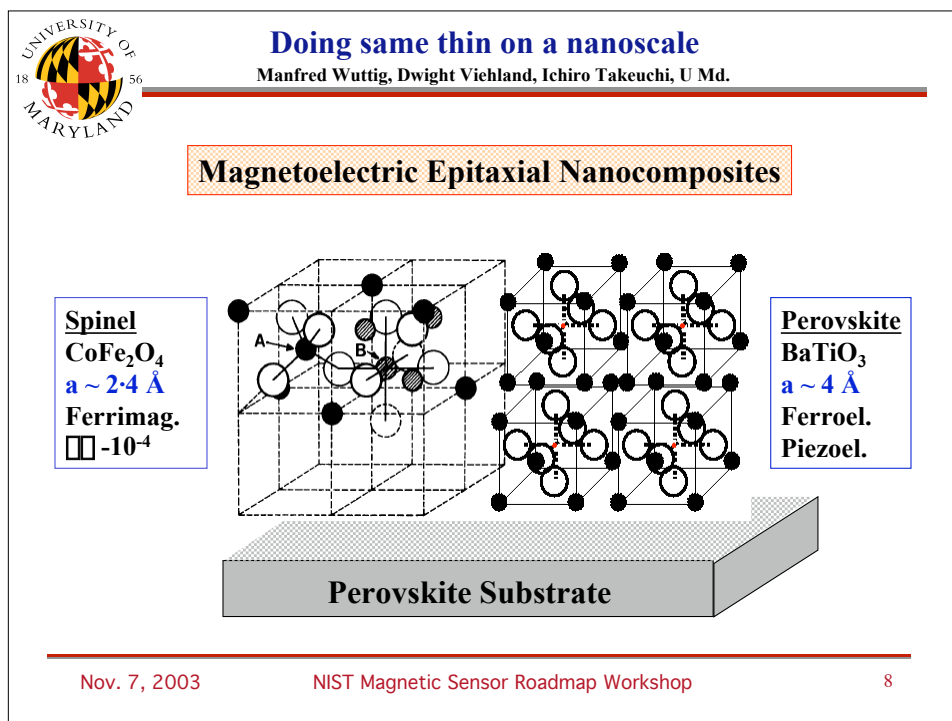
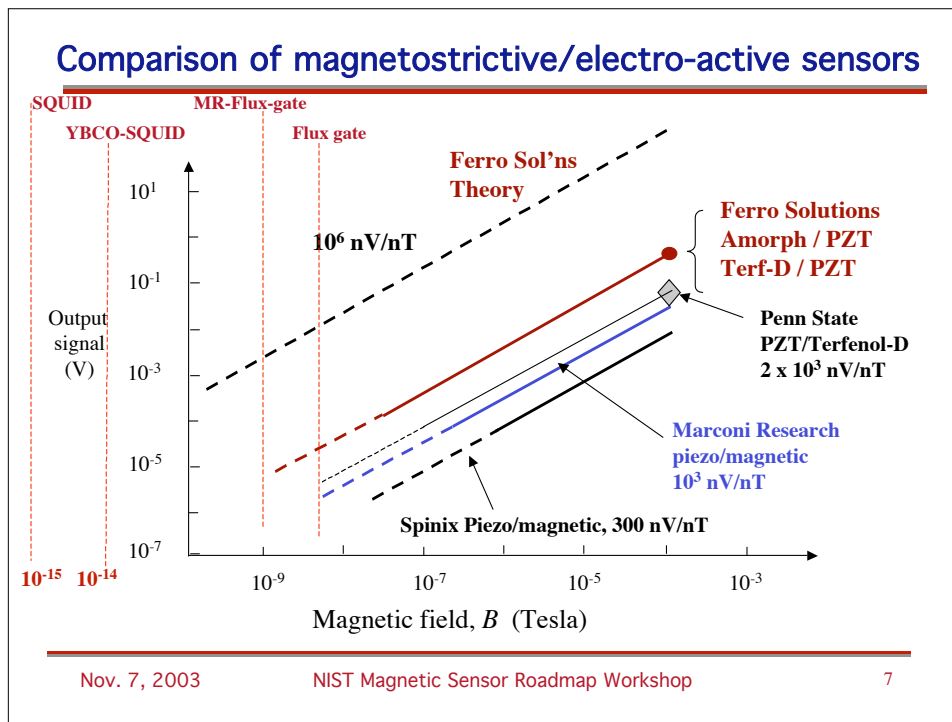
Can be AC biased: sensitivity = 6,000 nV/nT

J. K. Huang, R. C. O'Handley, Patent pending; SPIE Proc. March, 2003.

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MULTIFERROICS FOR SMART STRUCTURES AND DEVICES

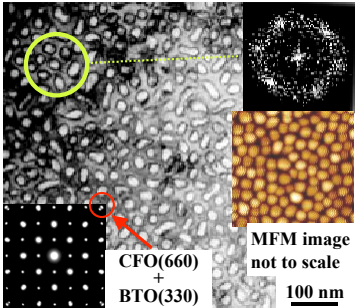
Manfred Wuttig, Dwight Viehland, Ichiro Takeuchi, U. Md

Promising Candidates

$\text{CoFe}_2\text{O}_4\text{-BaTiO}_3$
studied in bulk form 1976

Other systems:

$\text{CoFe}_2\text{O}_4\text{-FeBiO}_3$
 $\text{CoFe}_2\text{O}_4\text{-PbTiO}_3$



Self-assembled $\text{CoFe}_2\text{O}_4\text{-BaTiO}_3$
hexagonal ME hetero-nano-structure

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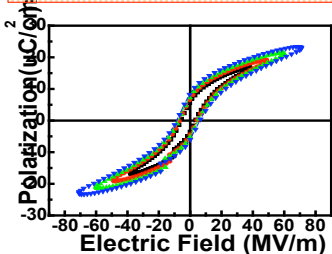
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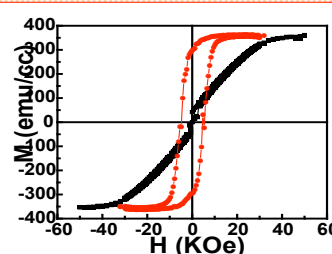


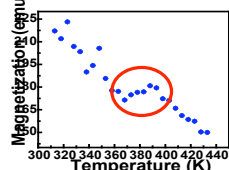
MULTIFERROICS FOR SMART STRUCTURES AND DEVICES

Manfred Wuttig, Dwight Viehland, Ichiro Takeuchi, U. Md

Magnetoelectricity of Epitaxial $\text{CoFe}_2\text{O}_4\text{-BaTiO}_3$ Nanocomposites



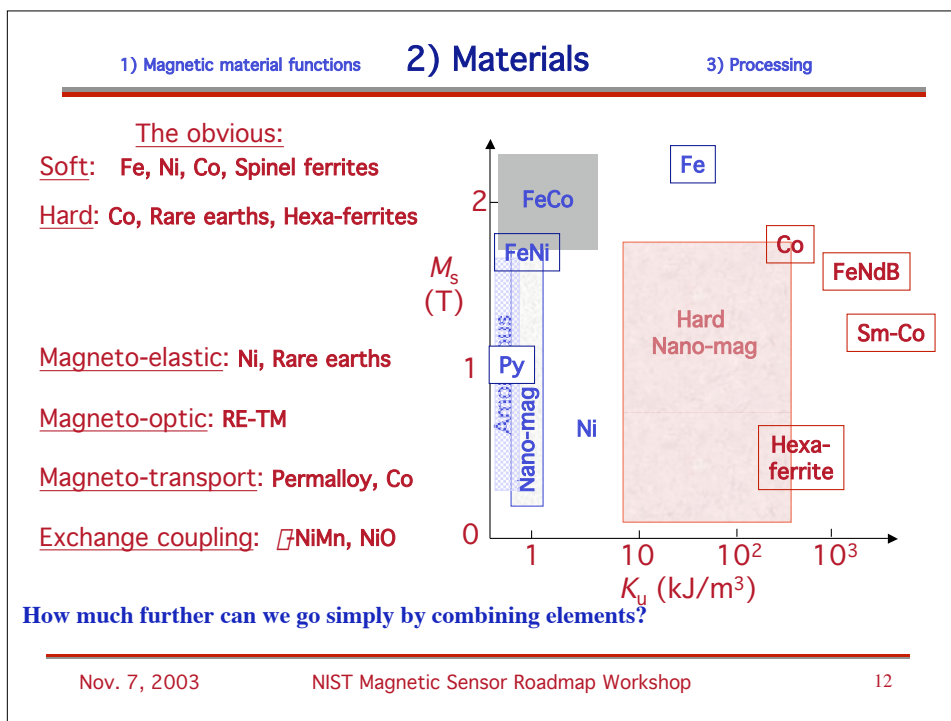
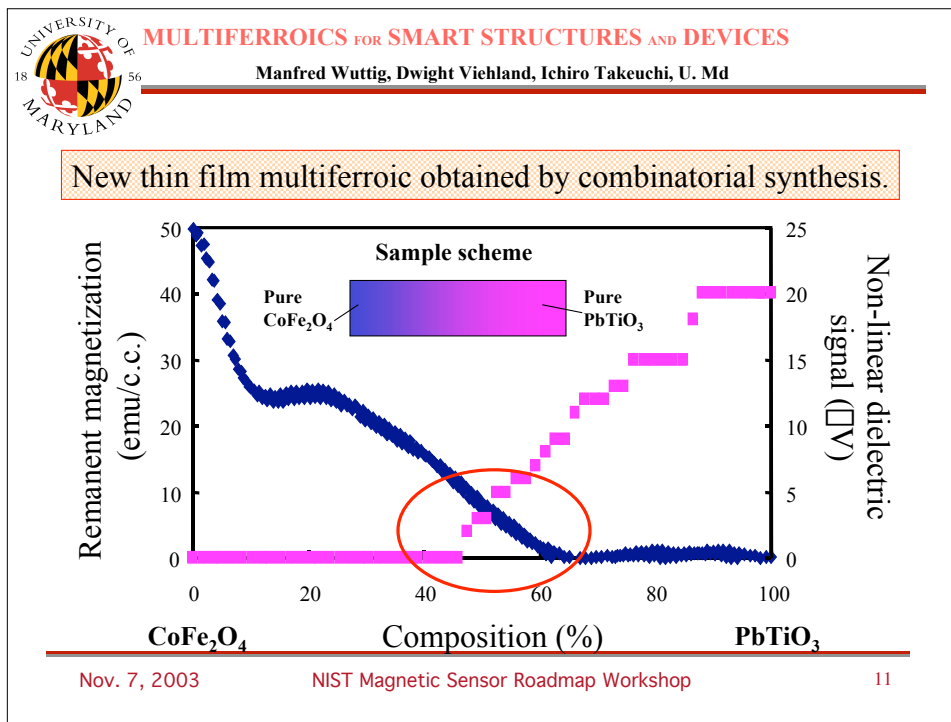




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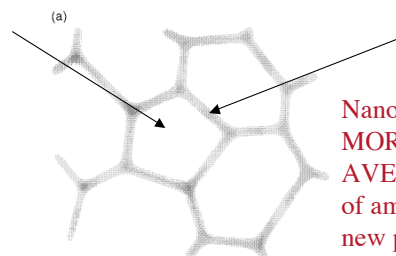
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Magnetic nano-structured materials

Crystalline properties

Amorphous properties



Nanostructured materials are MORE THAN JUST AVERAGE PROPERTIES of amorphous and crystalline materials; new property fields are accessible...

...and in nano-MAGNETIC materials, several competing length scales in play and can be exploited to advantage

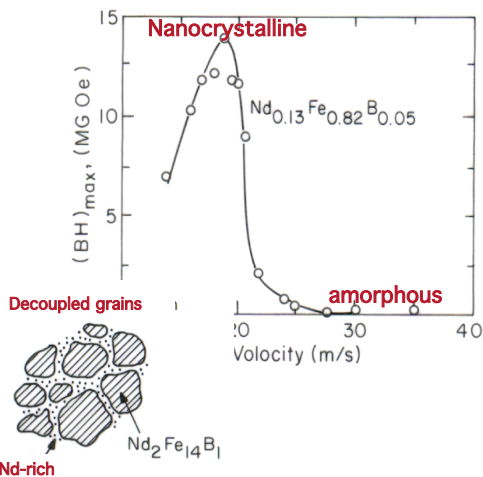
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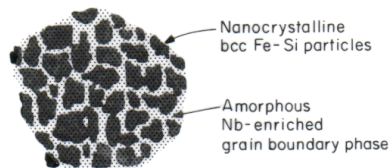
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Nano-magnetic materials - first pass

Rapidly solidified Nd-Fe-B
(Croat et al. '84)



Amorphous $\text{Fe}_{74}\text{Si}_{15}\text{B}_7\text{Cu}_1\text{Nb}_3$
□ "Fe₃Si" nano-crystals
in amorphous matrix
(Yoshizawa, '88)

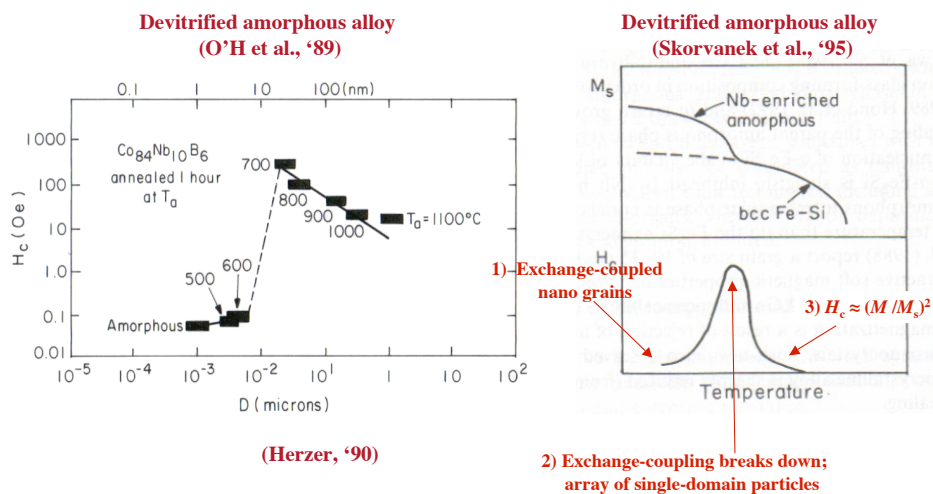


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Nano-magnetic materials - first pass

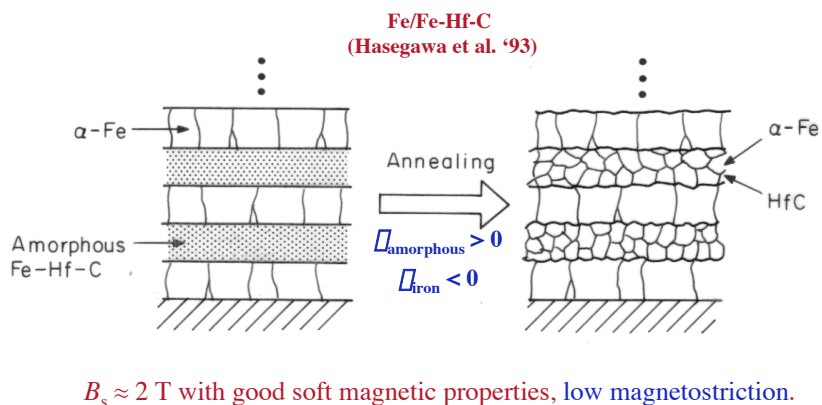


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Nano-magnetic materials - first pass



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Nano-magnetic materials - first pass

Crystallizing to α -Fe rather than Fe_3Si (DO_3) increases B_s :

Amorphous $\text{Fe}_{88}\text{Zr}_7\text{B}_4\text{Cu}_1$
 α -Fe nanocrystals
in amorphous matrix
(Suzuki et al., '91)

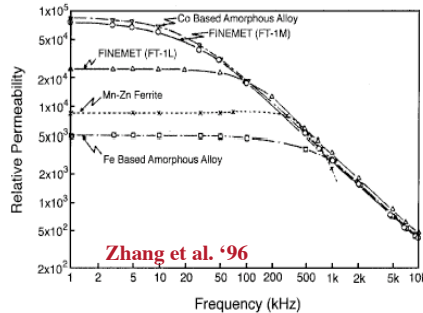
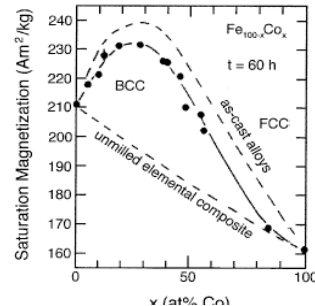


Fig. 67. Frequency dependence of the permeability for a FINEMET material as compared with amorphous material and with Mn-Zn ferrites [290,291].

Amorphous $\text{Fe}_{44}\text{Co}_{44}\text{Zr}_7\text{B}_4\text{Cu}_1$
 α -FeCo nanocrystals
in amorphous matrix (HITPERM)
(Willard et al., '98)



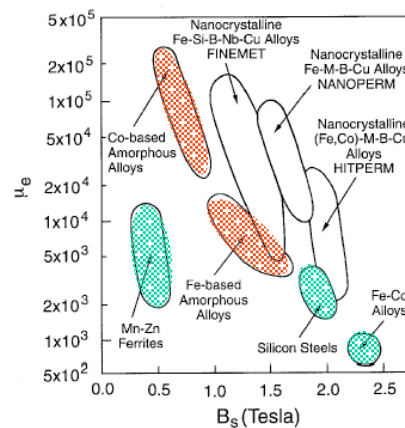
$B_s > 2$ T with permeability up to 4000
Now in thin film form: HITPERM/ SiO_2
McHenry group, C.M.U.

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Nano-magnetic material properties - first pass



(Machino et al., '95)

Fig. 1. Relationship between permeability, μ_e (at 1 kHz) and saturation polarization for soft magnetic materials (adapted from Ref. [193]).

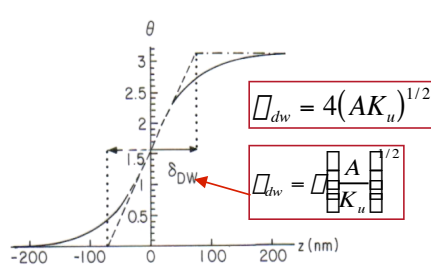
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Prior nano-magnetic results due to
Changed *length scales* => Completely new properties

Bloch domain walls (180°) walls in materials



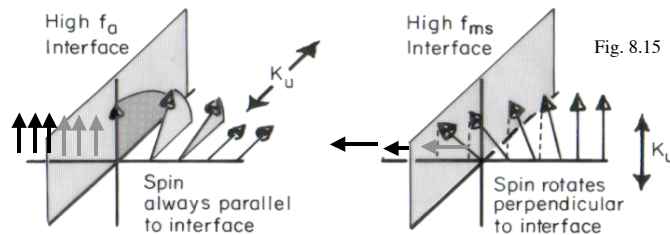
	K_u (kJ/m ³)	δ_{DW} (nm)	δ_{DW} (mJ/m ²)
Ni	4.5	150	0.85
Fe	48	45	2.8
Co	410	15.5	8.1
Fe ₁₄ Nd ₂ B	5,000	4.4	28
Amorphous	1	1,000	0.1

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Exchange lengths



M parallel to interface:

$$l_{ex}^{\parallel} = \frac{A}{K_u} = \frac{\delta_{DW}}{2}$$

M perpendicular to interface:

$$l_{ex}^{\perp} = \frac{A}{K_u + 2\mu_0 M^2} \text{ can be } \ll \delta_{DW}$$

Energy pushing spins back to EA

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Single-domain particles

For what spherical radius is magnetostatic energy smaller than domain wall energy?

$$\frac{1}{3} \mu_0 M_s^2 \frac{4}{3} r^3 \quad \mu_0 \gamma_{dw} \frac{4}{3} r^2 = 4 \mu_0 r^2 \sqrt{AK_u}$$

$$r_c = 9 \frac{(AK_u)^{1/2}}{\mu_0 M_s^2} \quad (\text{large } K_u)$$

$r_c = 8 \text{ nm for Co}, \quad 31 \text{ nm for Fe}_{14}\text{Nd}_2\text{B}$

Compare exchange energy cost to magnetostatic energy

$$r_c = \sqrt{\frac{9A}{\mu_0 M_s^2} \ln \frac{2r_c}{a}} \quad (\text{small } K_u)$$

$r_c \text{ for Fe (small } K) \approx 25 \text{ nm}$
(instead of $\approx 5 \text{ nm}$)

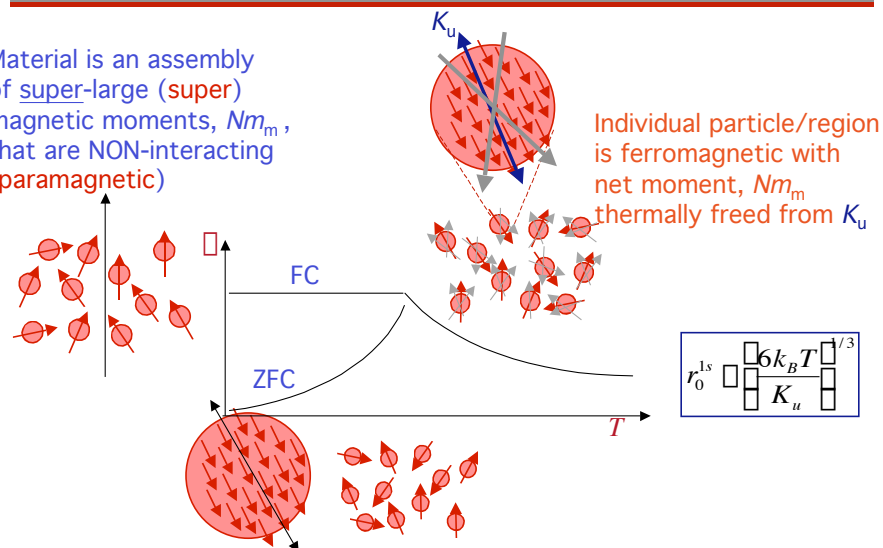
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Superparamagnetism

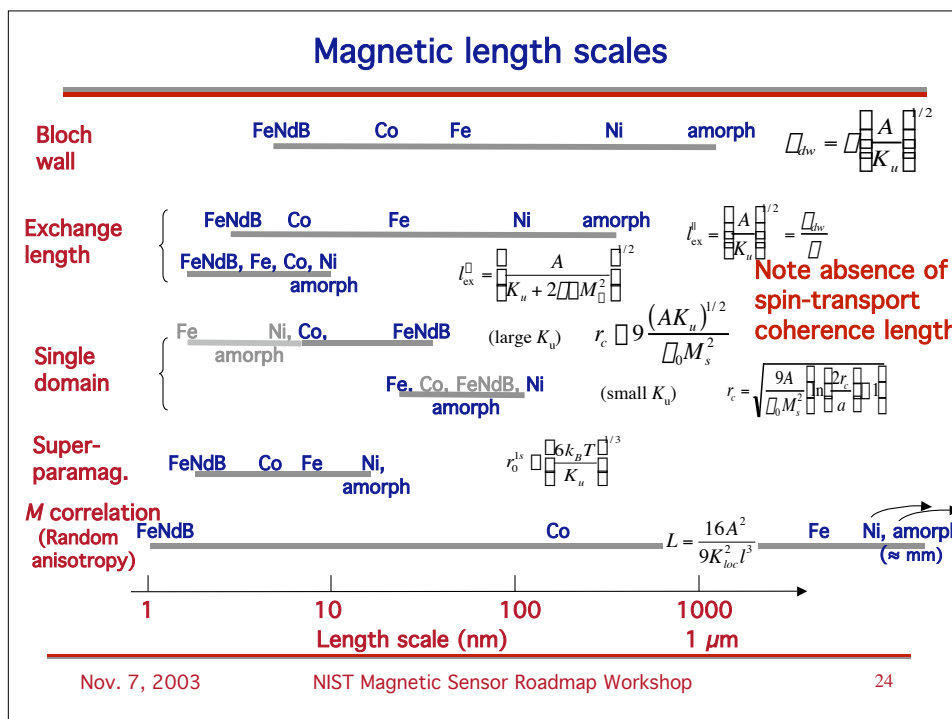
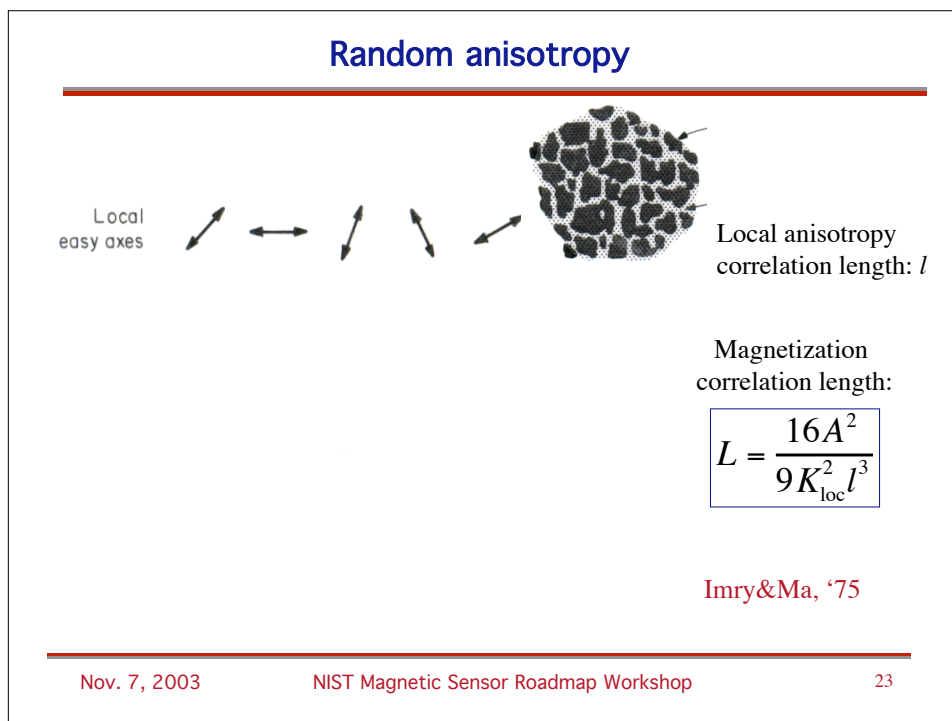
Material is an assembly of super-large (super) magnetic moments, Nm_m , that are NON-interacting (paramagnetic)



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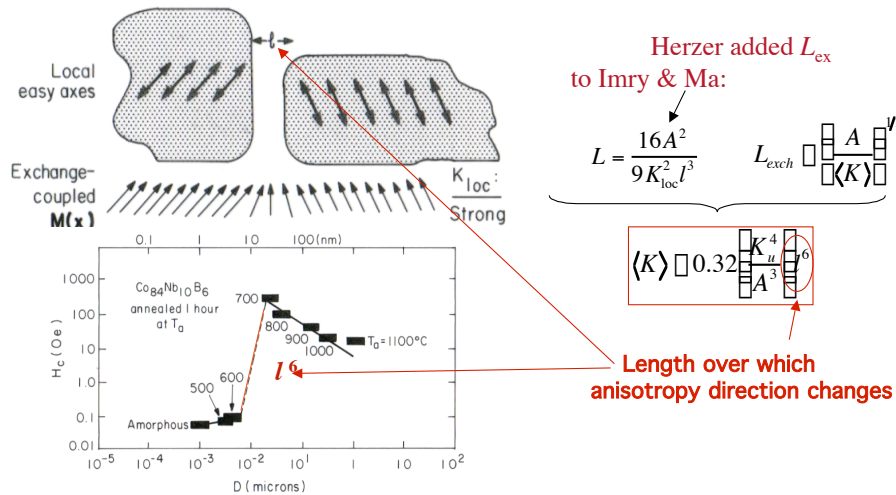
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Random anisotropy in nanocrystalline materials

Nano-magnetics - second pass: combine lengths

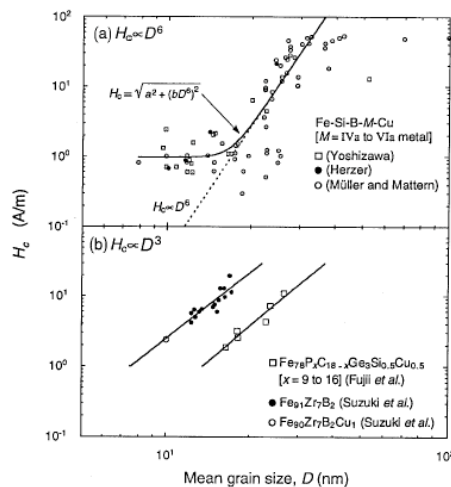


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Random anisotropy in nanocrystalline materials


Herzer. $H_c \propto D^6$

$H_c \propto D^3$ due to an additional, long-range, uniaxial anisotropy, possibly of magneto-elastic origin.

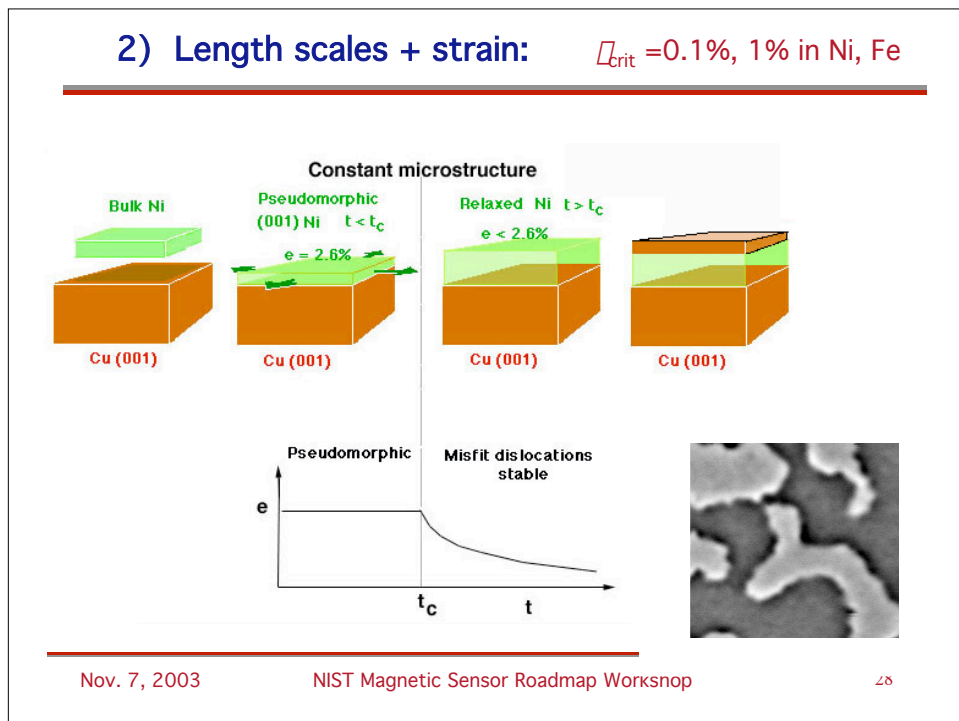
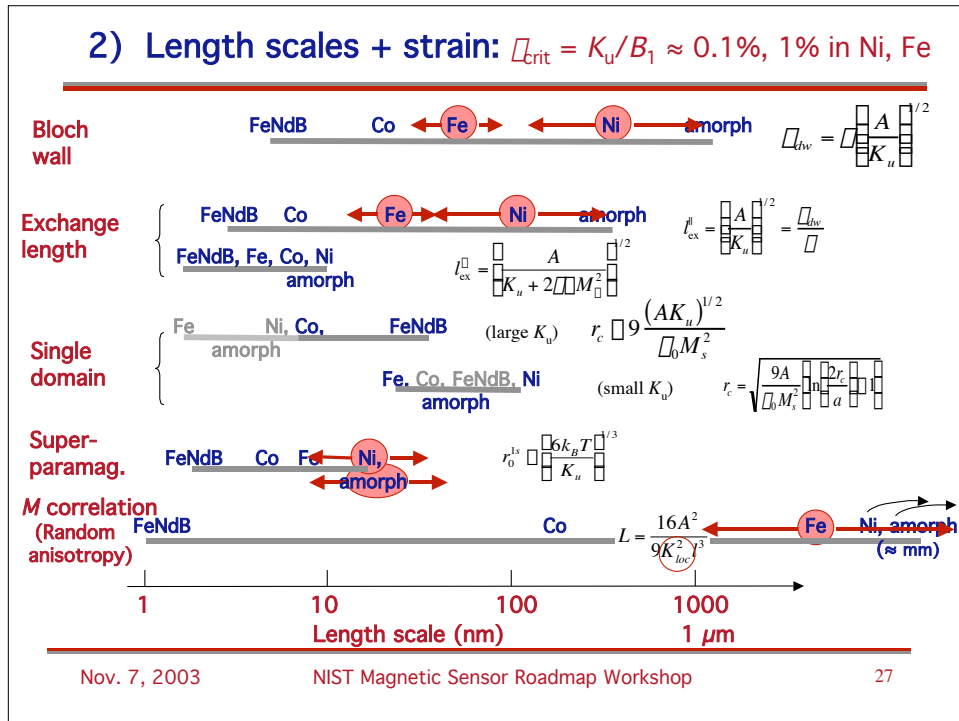
McHenry, Willard, Laughlin, Prog. Matls Sci. 44, 291-433 (1999)

Fig. 58. Coercivity, H_c , vs. grain size for (a) Fe-Si-B-TE-Cu (TE = early transition metal) (FINOMET) and (b) Fe-Zr-B-Cu (NANOPERM) and Fe-P-C-Ge-Si-Cu alloys [264-266].

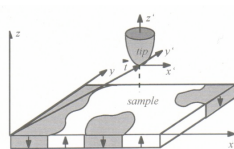
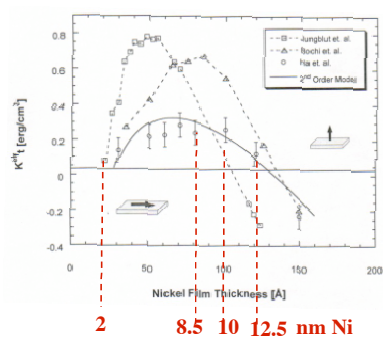
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2) Length scales + strain: $\epsilon_{crit} = 0.1\%$, 1% in Ni, Fe



Films strongly perpendicular 3 - 10 nm, mixed above 10 nm

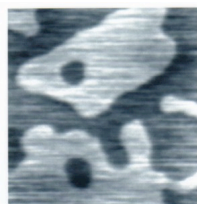


Fig. 1. Domain structure of 2 nm Ni/Cu over a 12 micron square.



Fig. 2. Domain structure of Cu/8.5 nm Ni/Cu over a 12 micron square.

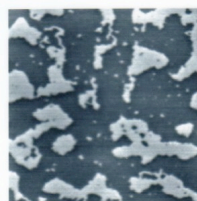


Fig. 3. Domain structure of Cu/10 nm Ni/Cu over a 12 micron square.

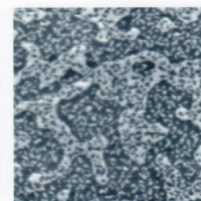


Fig. 4. Domain structure of Cu/12.5 nm Ni/Cu over a 12 micron square.

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